### 3.8.3 The Mark-Houwink-Sakurada relationship

Empirical relationship between the viscosity average MW and intrinsic viscosity.

 $[\eta] = KM_V^a \qquad (3.97)$ 

K, a= Constants for a particular polymer-solvent pair at a particular temperature

(See Table 3.10 for values of K and a)  $\langle M_w$  of sharp fraction of MWD is usually

substituted for M<sub>v</sub> since M<sub>v</sub> is difficult to measure directly>

#### Values of a

0.5≤a≤0.8 for random coil (0.5 @ Flory  $\Theta$  temperature ( $\alpha = 1$ ), 0.8 in thermodynamic good solvent ( $\alpha \propto M^{0.1}$ )→ $\alpha$  increase from M<sup>0</sup> to M<sup>0.1</sup>.

Polymer	Solvent	$T(^{\circ}C)$	$\mathbf{K} \times 10^{3a}$	$a^{b}$
cis-Polybutadiene	Benzene	30	33.7	0.715
it-Polypropylene	1-Chloronaphthalene	139	21.5	0.67
Poly(ethyl acrylate)	Acetone	25	51	0.59
Poly(methyl methacrylate)	Acetone	20	5.5	0.73
Poly(vinyl acetate)	Benzene	30	22	0.65
Polystyrene	Butanone	25	39	0.58
Polystyrene	Cyclohexane ( $\theta$ -solvent)	34.5	84.6	0.50
Polytetrahydrofuran	Toluene	28	25.1	0.78
Polytetrahydrofuran	Ethyl acetate hexane $(\theta$ -solvent)	31.8	206	0.49
Cellulose trinitrate	Acetone	25	6.93	0.91

Table 3.10 Selected intrinsic viscosity–molecular weight relationship,  $[\eta] = KM_v^a$  (77)

Source: J. Brandrup and E. H. Immergut, eds., Polymer Handbook, 2nd ed., Wiley, New York, 1975, sec. IV.

<sup>*a*</sup>European units, concentrations in g/ml. Units do not vary with *a*. Units of **K** are cm<sup>3</sup>·mol<sup>1/2</sup>/g<sup>1/1</sup>, <sup>*b*</sup> The quantity *a*, last column, is the exponent in equation (3.97).

Table 3.11 Values of the Mark-Houwink-Sakurada exponent a

a	Interpretation
0	Spheres
0.5-0.8	Random coils
1.0	Stiff coils
2.0	Rods

In general  $0 \le \alpha \le 2$  (See Table 3.11)

K is often given in terms of universal constant,  $\phi$ 

$$K = \phi \left(\frac{\overline{r_0^2}}{M}\right)^{3/2} \qquad (3.98)$$

 $r_0^2$ =Mean square of end-to-end distance of the unperturbed coil. Often MHS eq is conveniently expressed in the form of

$$[\eta] = \phi \left(\frac{\overline{r_0^2}}{M}\right)^{3/2} M^{1/2} \alpha^3 = K M^{1/2} \alpha^3$$
(3.99)

If the intrinsic viscosity is measured both in  $\Theta$ -solvent and good solvent, the expansion of coil is estimated as (Note  $\alpha$ =1 in  $\Theta$ solvent) < $R_e = R_{e0}\alpha$  (3.95)>

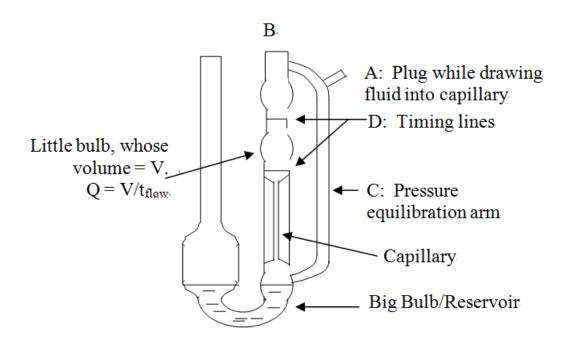
$$[\eta]/[\eta]_{\theta} = \alpha^3$$
 (3.100)

Values of  $\alpha$  vary from unity in Flory  $\Theta$ -solvent to about 2 or 3 increasing with M.

## 3.8.4 Intrinsic viscosity experiments

Mostly dilute solutions up to 1% polymer are made up. Then  $\eta_{rel} \sim 1.6$  for the highest concentration

**Ubbelohde viscometer** is used to measure the viscosity. (See below for the operation principle)



Capillary viscometer is working based on the Hagen-Poiseuille Law:

$$\begin{split} Q &= \frac{V}{t} = \frac{\pi R^4 \Delta P}{8\eta L} & (Q = \text{Volumetric flow rate}) \\ \text{Rearranging gives } (\Delta P = \rho g h) \\ &= \frac{\eta}{\rho} = \frac{\pi R^4 g h}{8 L V} t = K t & \text{K= Capillary constant} \\ &\to \eta = \text{Kt}\rho \end{split}$$

Then

$$\eta_r = \frac{\eta}{\eta_0} = \frac{t\rho}{t_0\rho_0},$$

For dilute solution  $\rho \sim \rho_o$ , then

$$\eta_{sp} = \eta_r - 1 = \frac{t - t_0}{t_0}.$$

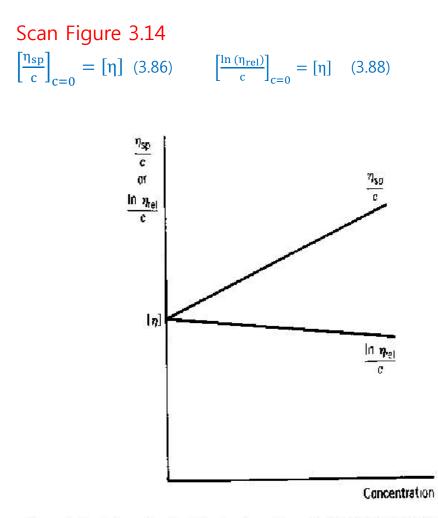


Figure 3.14 Schematic of a plot of  $\eta_{\rm ep}/c$  and in  $\eta_{\rm ep}/c$  versus c, and extrapolation to zero con-

### Two practical points must be noted:

- 1) Both lines must be extrapolated to the same intercept at zero concentration
- 2) Sum of the slopes is related through the Huggins eq  $\frac{\eta_{sp}}{c} = [\eta] + k'[\eta]^2 c \qquad (3.101)$

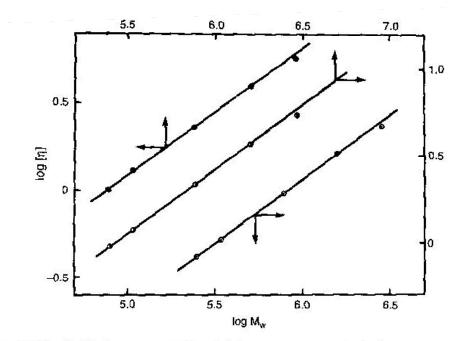
and through Craemer eq,  $\frac{\ln (\eta_{rel})}{c} = [\eta] - k'' [\eta]^2 c \qquad (3.102)$  Subtracting (102) from (101) gives

 $k' + k'' = 0.5 \tag{3.103}$ 

If either of these requirements is not met, molecular aggregation, ionic effects, or other problems may be indicated. For many polymer-solvent system k'=0.35, k''=0.15 (Negative).

In order to determine K and a in MHS eq, a double logarithmic plot of  $M_W$  vs intrinsic viscosity is prepared (Figure 3.15)  $[\eta] = KM_V^a$  (3.97)

For calibration, narrow MWD polymer should be used. Scan Figure 3.15 here



**Figure 3.15** Double logarithmmic plots of [n] versus  $M_w$  for anionically synthesized polystyrenes, which were then fractionated leading to values of  $M_w/M_n$  of less than 1.06. Filled circles in benzene, half-filled circles in toluene, and open circles in dichloroethylene, all at 30°C (75). The arrows indicate the axes to be used. Units for [n] in 100 mi/g.

# 3.8.5 Example Calculation Involving Intrinsic Viscosity

Say we are interested in a fast, approximate molecular weight of a polystyrene sample. We dissolve 0.10 g of the polymer in 100 ml of butanone and measure the flow times at 25°C in an Ubbelhode capillary viscometer. The results are

Pure butanone	110 s	
0.10% Polystyrene solution	140 s	

Starting with equation (3.84), and nothing that the flow time is proportional to the viscosity,

$$\eta_{\rm rel} = \frac{\eta}{\eta_0} = \frac{140}{110} = 1.273$$
$$\eta_{\rm sp} = \eta_{\rm rel} - 1 = 0.273$$
$$\frac{\eta_{\rm sp}}{c} = \frac{0.273}{0.001} \,\mathrm{ml/g} = 2.73 \times 10^2 \,\mathrm{ml/g}$$

As an approximation, assume that the concentration is near zero, and the [n]

 $2.73 \times 10^2$  ml/g, equation (3.86). This obviates the extrapolation in Figure 3.14 that is required for more accurate results. Using the Mark-Houwink-Sukurada relation, equation (3.97) and Table 3.11, we have

$$[\eta] = \mathbf{K} M_V^a$$
  
2.73×10<sup>2</sup> = 39×10<sup>-3</sup> M\_V^{0.58}  
 $M_V = 4.26 \times 10^6$  g/mol

Note that the units of K are irregular, depending on the value of a. Usually the units of K are omitted from tables.

#### 3.9 Gel permeation chromatography

Also called size exclusion chromatography (SEC)

Depending on the hydrodynamic radius, molecule may or may not enter small pores in a bed of cross-linked polymer (Stationary phase). Smaller molecules diffuse in and out of the pores via Brownian motion and delayed. Large molecules pass by and continue in the mobile phase. (See Figure 3.16)