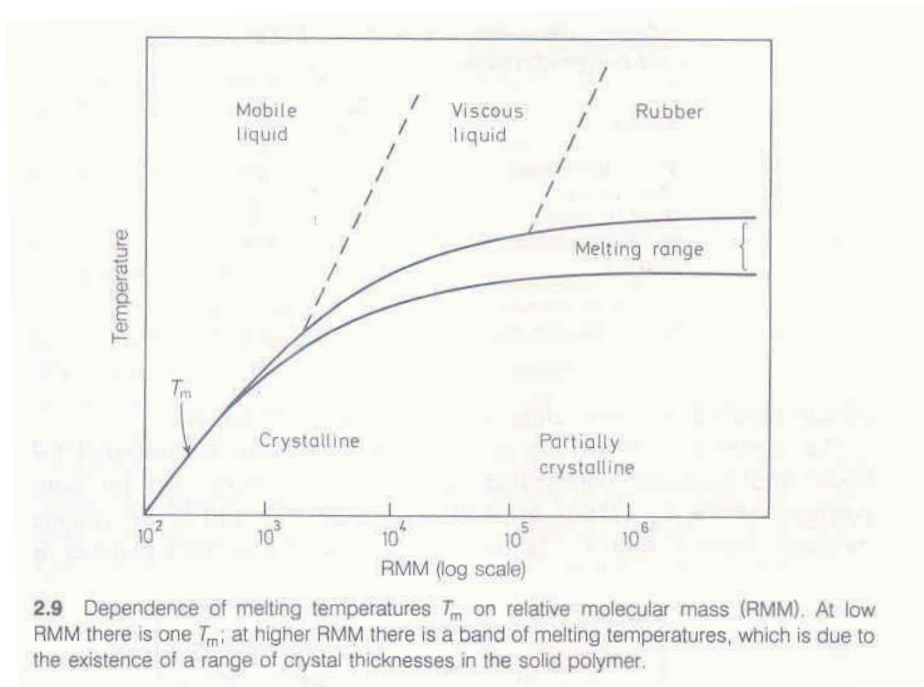


2.5 Crystallization and melting (Continued)

Mw vs Tm (Figure 2.9)



For $M_w < 10^3$

- ▶ A sharp single melting peak.
- ▶ T_m increases with M_w .

For $M_w > 10^3$

- ▶ T_m levels off (asymptotic) with melting range (Due to the MWD)
- ▶ $T_m \propto$ lamella thickness
- ▶ Large molecule, high structural regularity, crystallization at high temperature gives thick lamella.
- ▶ T_m is defined for the thickest, largest and most perfect crystal obtainable. (See a melting endotherm from DSC measurements)

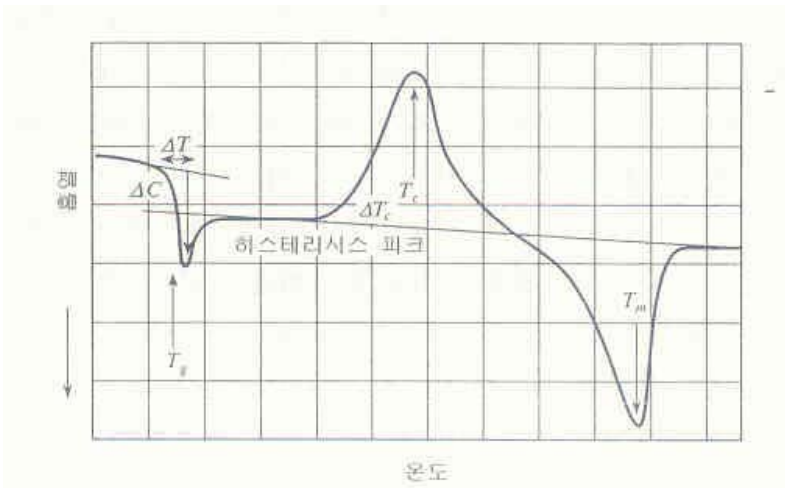


Figure Typical DSC measurements.

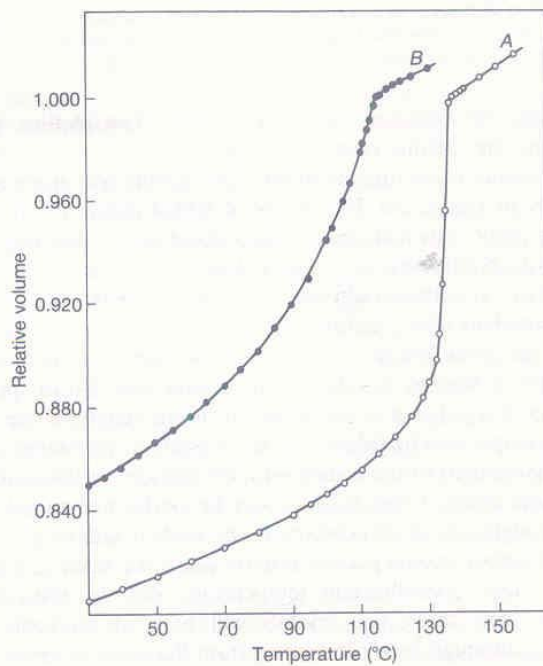
Table 2.1 for T_m and T_1

Table 2.1 The melting temperature T_m (°C) of some polymer crystals

Polymer	T_m (°C)	T_1 (°C)
Linear polyethylene	138	56
Polypropylene	176	86
Polyoxymethylene	180	90
Poly(ethylene terephthalate)	266	158
Nylon 6.6	264	157

T_1 is the temperature of maximum crystallization rate deduced from the approximate empirical rule given in eqn 2.3.

Figure 2.10 Relative volume (1/density) during heating for BPE (B) and LPE (A)



2.10 Plot of relative volume during heating against temperature T . The relative volume is the ratio of the specific volumes at T and T_m . Curve A is for linear polyethylene; curve B is for branched polyethylene (after Mandelkern).

LPE= sharp melting

BPE= broad melting

In molten state, there is no difference between LPE and BPE.

► Two steps in melt crystallization = Nucleation + Growth

▷ At high temperature: slow nucleation + rapid growth (low viscosity)

→ thick crystal → high T_m

▷ At low temperature: fast nucleation + slow growth (high viscosity)

→ thin crystal → low T_m

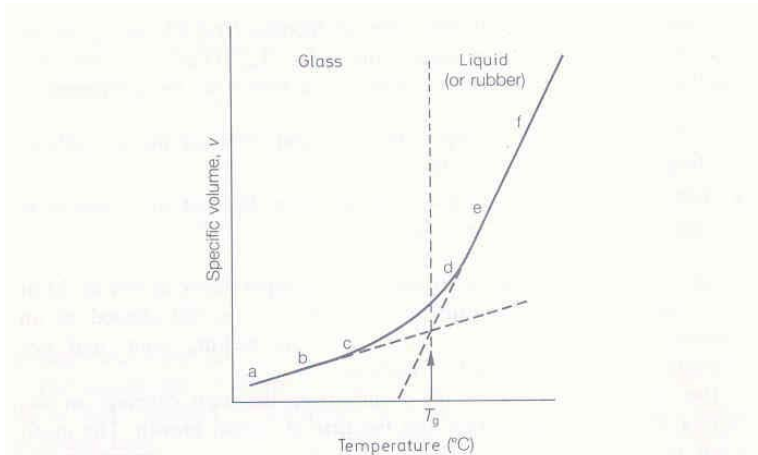
► Maximum rate of crystallization occurs at a degree of supercooling of approximately

$$\Delta T = T_m - T_c = 0.2 T_m \quad (\text{K}) \text{ or}$$

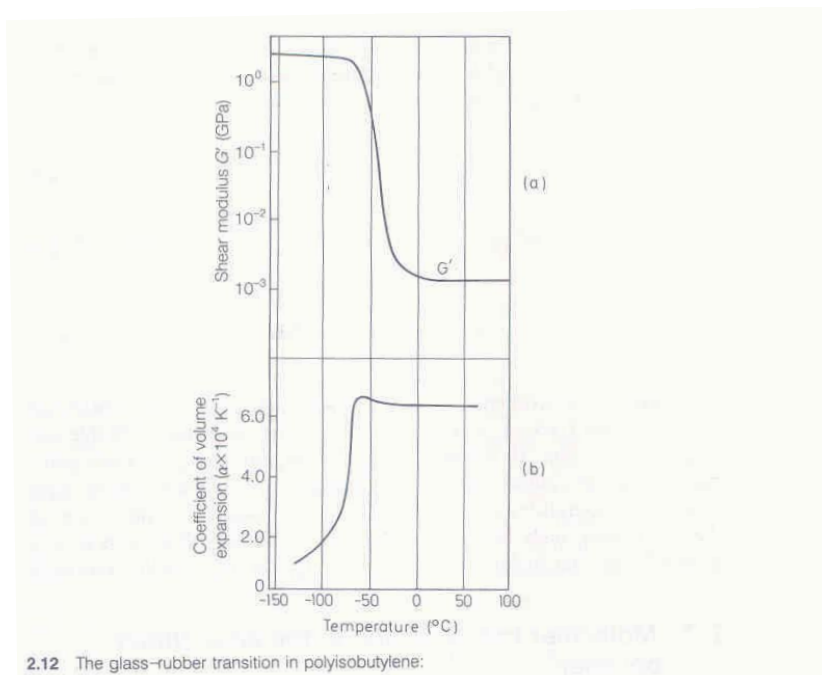
$$T_1 = 0.8 T_m \quad (2.8)$$

2.6 Glass Transition Temperature

See [Figures 2.11 and 2.12](#) for Specific volume–Temperature and Modulus–Temperature.



2.11 The glass transition temperature T_g is obtained from an experiment in which the specific volume v is measured whilst the specimen is cooled at a fixed rate, usually 1°C per minute. The construction for obtaining T_g from the data is shown. For polymer of low RMM, T_g marks the transition from glass to liquid; for high RMM, T_g marks the transition from glass to rubber.



2.12 The glass–rubber transition in polyisobutylene:

►In glassy state:

▷As $T \uparrow \rightarrow$ Secondary force (intermolecular) $\downarrow \rightarrow$ Separation of polymer chains \uparrow (Thermal expansion) \rightarrow Density \downarrow (Specific volume \uparrow) (See secondary force vs thermal energy in Lecture # 187))

-----Note from 187

Chemical Bond

{Primary bond (covalent bond)
{Secondary bond

Van der Waals < dipole < H-bond < ionic bond

Table 1.1 Dissociation energy and length of different bonds

Bond type	Energy (kJ mol ⁻¹)	Bond length (nm)
Covalent bond	300-500	0.15 (C-C; C-N; C-O) 0.11 (C-H) 0.135 (C=C)
van der Waals bond	10	0.4
Dipole-dipole bond	> 10	0.4
Hydrogen bond	10-50	0.3

Compare the above bond energy with thermal energy(RT)

$$RT = 2,5\text{kJ/mole @ } 300\text{K}$$

$$= 4 \text{ kJ/mole @ } 500\text{K}$$

$$(R=1,987\text{cal/mole K})$$

If thermal energy > bond energy \Rightarrow Bonds tend to dissociate.

At melting

Covalent bondings are preserved.

Rupture & reestablishment of secondary bonds occur.

----- End of 187

▷no segmental motion, no conformation change \rightarrow Thermal expansion coefficient (α , β) is constant and small. (See Figure 2.12)

$$\alpha = (1/L)(dL/dT), \quad \beta = (1/V)(dV/dT) \quad @ \quad P=\text{constant}$$

▷ Only rotation and vibration of elements and groups exist.

▶ Around T_g , segmental motion (jump) starts → Expansion coefficient increases with T (See increasing slope in Figure 2.11, and a large drop in modulus in Figure 2.12).

▶ In melt, $>10^6/s$ segmental motions!!! (See constant and large slope in Figure 2.11)

(One segment typically consists of 20–30 repeat units.)

Measurement of T_g

i) Specific volume measurement (See Figure 2.11)

ii) DSC (Differential Scanning Calorimeter)

T_g = point of inflection (Second order transition) (Endothermic)

T_m = endothermic peak,

T_c (Exothermic peak) (See a typical DSC curve)

▶ T_g depends on the rate of cooling.

Slower cooling → lower T_g

(Prolonged process time vs constant relaxation time of polymer chain → Viscous (flow) rather than elastic (recovery) nature is imposed on the polymer. $\gamma = \gamma_0 \sin \omega t$)

iii) Dynamic mechanical method (DMTA)

▶ Heat up the sample at a constant frequency strain ($\gamma = \gamma_0 \sin \omega t$).

▶ The temperature at which the external frequency is matched with that of segmental motion, a function of temperature, is T_g .

▶ At T_g , elastic energy is tremendously damped, dissipated (See the modulus drop and $\tan \delta$ peak at T_g).